

Reactivity Assessment of Bauxite Residue-Rich Slags

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Abstract

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Although multiple valorisation routes have been proposed and patented for bauxite residue, BR, its use is currently limited to less than 3 % of the annual >150 million tonnes being produced. One of the reasons is BR's low reactivity in a cementitious environment. In this paper, a method for increasing the reactivity of bauxite residue after vitrification is used and the resulted produced slags are tested as inorganic polymer precursors. Four different slags were prepared by mixing varying amounts of BR, clay bricks (CB), and fly ash (FA). The mixtures were melted at 1375°C and then quenched in water. The slags were milled to reach a similar Blaine value of about 5000 cm²/g. The reactivity of the binders, associated with the polymerization degree of the glass phase, was assessed by monitoring the heat flow. Results demonstrate highest reactivity for Slag 4, followed by Slag 1, 2 and 3. The fresh inorganic polymer (IP) pastes were cast in an ultrasonic device to estimate the evolution in mechanical properties and the setting time at 20°C and 65 % relative humidity. The fastest slag to set was Mix 4 after about 17 minutes, followed by Slag 1, 2 and Mix 3, all after about 2.5 h. Inorganic polymer samples were also produced, and their compressive strength was measured for 1, 7, 28 and 90 days. The strongest sample was made with Slag 2 (75 BR, 12.5 FA and 12.5 CB wt %) with a compressive strength of more than 100 MPa after 90 days. Slag 3, with the lowest amount of CaO and lowest Al/Si ratio, had the lowest mechanical performance.

Keywords: Bauxite residue, Iron rich slag, Calorimetry, Reactivity assessment, Inorganic polymers.

1. Introduction

The highly alkaline residue, BR, is generated in the Bayer process of alumina production. About 1 to 1.5 tonnes of the residue are produced per ton of alumina, depending mainly on the nature of the bauxite ore and the process parameters [1]. Although multiple valorisation routes for use as cementitious material have been proposed and patented for BR, its use is currently limited to less than 3 % of the annual >150 million tonnes being produced. One of the reasons is bauxites residue's low reactivity in a cementitious environment. In this paper, a method for increasing the reactivity of bauxite residue after vitrification is used and the produced slags are tested as inorganic polymer precursors.

Generally, the use of inorganic polymers as an alternative binder drew the attention because of their excellent properties, such as high compressive strength, resistance to acidic solutions and

thermal stability [2, 3], in combination with their relatively lower CO₂ footprint [4]. Iron-rich precursors, for example copper [5], lead [6] or ferro-nickel slags [7] have shown already a potential for the synthesis of inorganic polymers and the re-use of industrial wastes. Characteristics of these precursors include among others, a partially vitrified structure and ferrous iron [8]. Chemically considered, these slags do not differ significantly from BR, apart from the mineralogy and the oxidation state of iron. This led to the assumption that a chemical adaptation followed by thermal processing can turn BR into a suitable precursor for IP, as has been shown by others [9, 10] that the process of transforming the BR slag into a reactive slag precursor is feasible.

In this paper four different mixtures will be produced based on bauxite residue, mixed with other residual resources, namely clay bricks and fly ash. Clay bricks contain a substantial amount of aluminosilicates, while especially aluminum and silicon are lacking in BR, thus addition of clay bricks could help to increase these. Fly ash can bring in some calcium, which is known to increase the reaction rate. These mixtures will be molten and quenched to obtain an amorphous slag, as it is known that the degree of reactivity is related to the amorphous fraction. The mixtures will then be tested on their reactivity and mechanical properties. Although this procedure consumes a lot of energy, needed for melting the ingredients, it has the big advantage that a mixture can be obtained with constant quality. For certain applications this is of utmost importance.

2. Materials and Methods

Four different slag chemistries were prepared by melting BR in combination with other residual streams. The aim was to increase the SiO₂ and CaO content in the BR in order to optimise the fraction of molten material and to obtain a more amorphous and reactive glass upon quenching [11, 12]. The addition of SiO₂ was in the form of milled clay bricks, rich in SiO₂ (74.1 wt %) and Al₂O₃ (13.5 wt %). To increase the CaO content of the slag, a calcium rich fly ash was used, i.e., CaO content of 50.2 wt %. To increase the molten phase in the liquid slags, which necessitates Fe²⁺ for the chosen temperatures, carbon was added in the form of carbon black. The as-received wastes and the slags were characterized before use. The mineralogical composition was evaluated by X-ray diffraction (XRD) recorded using a $\theta/2\theta$ diffractometer (PANalytical X'Pert Pro, NL), equipped with a fast RTMS detector (PIXcel 1D, PANalytical), with Cu K radiation (45 kV and 40 mA, 5-80 °2 θ range, with a step scan of 0.02 °2 θ , and time per step of 200 s). Full quantitative phase analyses (FQPA) were assessed for quantifying both crystalline and amorphous phases in the slags, using the combined Rietveld-reference intensity ratio methods, using 10 wt % of α -Al₂O₃ (NIST SRM 676a) as an internal standard. The sample's chemical composition was determined by X-ray fluorescence spectroscopy (Philips X'Pert PRO MPD spectrometer). The amount of carbon black added in the mixtures was calculated as 1.5 wt % of the BR content. Before melting, the different raw materials were blended as a slurry using a Hobart mixer, ensuring homogeneous mixtures. Subsequently, the slurry was dried at 60 °C for 48 h. The obtained dry mixtures were then melted in an induction furnace at 1375 °C using a steel crucible. The atmosphere in the furnace during melting was maintained at 40/20 CO/CO₂ (l/h) to ensure a reducing environment, further promoting the formation of a liquid phase. The molten slags were quenched in water after one hour residence in the furnace at 1375 °C. The rapid cooling of the molten slag results in an amorphous and thus more reactive material. The wet slag was subsequently dried at 100 °C for 48 h. The samples were then milled in a ball mill for 4 to 5 h, depending on the hardness of each sample. The density of the slags was measured with an Ultrapys 5000 Micro, Anton Paar. The reactivity of the slags with a Na-silicate activating solution (molar ratio of SiO₂/Na₂O of 2.0 and H₂O content of 65 wt %) was investigated by isothermal heat flow calorimetry of paste samples recorded at 20.0±0.2 °C for 72h using a TAM Air calorimeter (TA Instruments). The samples were internally stirred using the Admix Ampoules in order to monitor the initial heat flow and the initial reactions. The liquid to solid (L/S) was kept constant to 0.45. The dynamic E-modulus (E-mod) development was measured with an ultrasonic

3.3 Mechanical Properties of the Produced Slags

The compressive strength data are shown in Figure 4. At the ages of 1 and 7 days, the inorganic polymer prepared with Slag 4 shows the highest strength values, followed by Slag 1, Slag 2 and Slag 3, respectively. After 28 days, this sequence is not observable anymore. At later ages, Slag 1, 2 and 4 perform similarly. Slag 3 has overall the lowest strengths results. This is the slag with the lowest CaO content and Al₂O₃/SiO₂ ratio.

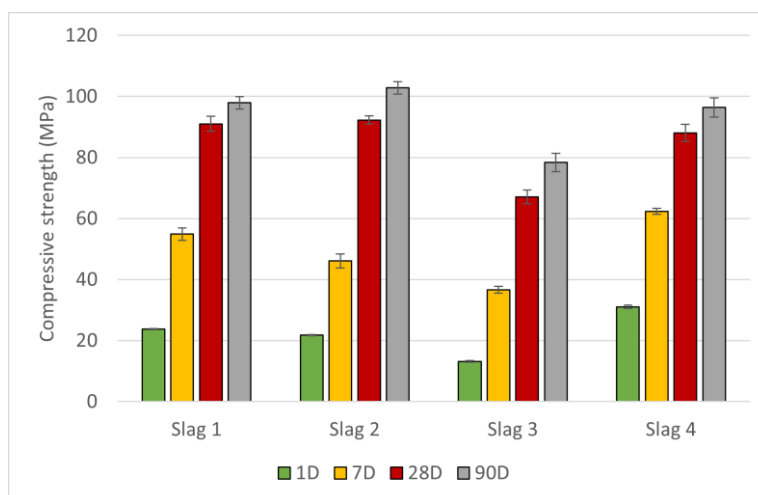


Figure 4. Compressive strength values at 1, 7, 28 and 90 days.

4. Conclusion

Inorganic polymer samples were prepared with treated BR as the main component. The reactivity of the binders, associated with the polymerization degree of the glass phase, was assessed by monitoring the heat flow and E-modulus development. There is a possible correlation between the E-modulus value at 24h and the cumulative heat flow at the same time, results that could be used as a way to deduce qualitative information for E-modulus via calorimetry measurements. Also, there might be a correlation between the cumulative heat release at 24 hours and the compressive strength values at the same time. Inorganic polymer samples were also produced, and their compressive strength was measured for 1, 7, 28 and 90 days. The strongest sample was made with Slag 2 (75 BR, 12.5 FA and 12,5 CB wt %) with a compressive strength of more than 100 MPa after 90 days.

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